PAPER • OPEN ACCESS

Multi-component engineering to enable long-term operational stability of perovskite solar cells

To cite this article: Haibing Xie and Monica Lira-Cantu 2020 J. Phys. Energy 2 024008

View the article online for updates and enhancements.

You may also like

- Printable materials for printed perovskite solar cells
 Wangnan Li, Zhicheng Zhong, Fuzhi Huang et al.
- Roadmap on commercialization of metal halide perovskite photovoltaics Shien-Ping Feng, Yuanhang Cheng, Hin-Lap Yip et al.
- <u>Surface modulation of halide perovskite</u> <u>films for efficient and stable solar cells</u> Qinxuan Dai, , Chao Luo et al.

Journal of Physics: Energy

PAPER

OPEN ACCESS

CrossMark

RECEIVED 1 November 2019

REVISED

10 February 2020

ACCEPTED FOR PUBLICATION 23 March 2020

PUBLISHED 24 April 2020

Original content from this work may be used under the terms of the Creative Commons Attribution 4.0 licence.

Any further distribution of this work must maintain attribution to the author(s) and the title of the work, journal citation and DOI.



Multi-component engineering to enable long-term operational stability of perovskite solar cells

Haibing Xie D and Monica Lira-Cantu

Catalan Institute of Nanoscience and Nanotechnology (ICN2), CSIC and the Barcelona Institute of Science and Technology (BIST), Building ICN2, Campus UAB, Bellaterra, E-08193 Barcelona, Spain

E-mail: xhaibing@gmail.com and monica.lira@icn2.cat

Keywords: perovskite solar cells, long-term operational stability, multi-component engineering

Abstract

With a record efficiency above 25%, the main hurdle for the commercialization of perovskite solar cells (PSCs) is their long-term operational stability. Although different strategies have been applied, the stability of PSCs is still far below the 25 year requirement demonstrated by commercial photovoltaic technologies. To advance in the former, a lab-scale stability analysis should resemble real testing conditions, and this is only possible through the interaction of several stress factors. Here, we briefly introduce the reader to the general degradation mechanisms observed on PSCs and the state-of-the-art strategies applied to realize long-term stable devices. Finally, we highlight the imperative need to engineer multiple components of the PSCs simultaneously and propose a rational design of PSC's constituents to obtain long-term operational solar cells. This perspective article will benefit the progression of PSCs as a reliable photovoltaic technology.

1. Introduction

Hybrid organic-inorganic metal halide perovskites, with a representative perovskite formula of ABX₃ (A = methylammonium (MA), formamidinium (FA), Cs; B = Pb, Sn; X = I, Br), have achieved a certified solar conversion efficiency of 25.2%, approaching the record efficiency of single-crystal silicon solar cells of 26.1% [1]. Nevertheless, their reported lifetime ranges from minutes to around one year [2], which is far from the benchmark of 25 years achieved by silicon solar cells. Commercial photovoltaic cells and modules must comply with several industry-standard qualification tests described in the International Electrotechnical Commission (IEC) standards on terrestrial photovoltaic modules (including the IEC 61215, IEC 61730 and IEC 61646) [3]. These reliability PV standard tests comprise conditions that are currently severe for PSCs, e.g. damp-heat test (85% RH and 85 °C for 1000 h), UV preconditioning test and thermal cycling test (-40 °C to 85 °C, 200 cycles); however, as PSC technology is still in its early stage of development, it is not ready yet for standardization. For this reason, the international photovoltaic research community has proposed and developed reliability protocols useful to validate and to compare the PSC device lifetime among different laboratories [4-6]. Among them, the ISOS protocols [7], which were recently updated for PSC reliability tests [6], include different stress levels that increase in intricacy comprising from basic lab-scale research to industry device testing. We have observed that the most applied stability tests published for PSCs relate to less stressful conditions, e.g. analysis in the dark, illumination without UV, under inert gas, at room temperature or short testing times. Also rare are stability studies that are comparable to real testing conditions (e.g. outdoor analysis), especially applying more than one stress factor simultaneously (e.g. under continuous light irradiation, temperature over 65 °C and bias voltage). This limits the advancement of the technology towards commercial stable PSCs.

In this perspective article, we highlight the importance of performing stability tests that incorporate multiple stress factors in a single analysis. Most importantly, we show that the path to stable PSCs comprises the simultaneous engineering of multiple constituent materials and interfaces of the device. Thus, we begin by briefly discussing the degradation mechanisms of PSCs under five relevant stress factors: humidity, oxygen, light, heat and bias voltage and their combinations. We also include a concise overview of the

existing strategies currently employed for PSC stability enhancement. Finally, we propose a comprehensive strategy to improve the durability of PSCs based on the engineering of multiple PSC components simultaneously, addressing each of their compositional layers of the device: absorber (perovskite), transporting layers, electrodes and the corresponding interfaces.

2. Degradation mechanisms and stress factors

The aim of this section is not to review the detailed state-of-the-art related to the degradation mechanisms of PSCs. To do so, the reader can be redirected to the extensive work and literature reviews recently published in the area [8–10]. Our objective is to briefly introduce the reader to the main stress factors that affect PSC stability and the related degradation mechanisms, in an attempt to facilitate the subsequent discussions on strategies for stability enhancement. Stress factors referred to here are humidity, oxygen, light irradiation, temperature (heat) and bias voltage. The study of PSC stability by a single stress factor contributes to the understanding of degradation mechanisms occurring specifically in any of the constituent materials; however, the application of multiple stress factors in parallel can closely replicate the real operational conditions of the PSCs and thus can result in a more realistic stability analysis.

2.1. Single stress factors

The testing atmosphere, especially the presence of humidity and oxygen, affects PSC device stability. Due to the hygroscopic nature of the amine groups present in the organic MA and FA cations, humidity can lead to the hydrolysis of the halide metal perovskite into hydrated products, such as $MA_4PbI_6 \cdot 2H_2O$ and PbI_2 [11, 12]. In contrast, the halide perovskite under a pure oxygen atmosphere does not show significant degradation [13]. In commercialized modules, solar cells are well encapsulated to greatly avoid the erosion of humidity and oxygen.

For PSCs, UV light (< 400 nm) is by far the most detrimental wavelength range of the entire AM 1.5 G solar spectrum. The degradation mechanisms taking place are mainly related to the degradation of transporting layers and their interfaces with perovskite. TiO₂ based PSCs experience instability under UV illumination due to the deep trapping of electrons caused by light-induced desorption of surface-adsorbed oxygen or the decomposition of perovskite caused by the photocatalytic effect of TiO₂ at the interfaces [14, 15]. This light-induced degradation is characteristic of many other metal oxides under UV light, such as ZnO electron transporting layers (ETL) [16]. Recent studies suggest the application of materials that are insensitive to UV light to mitigate this issue [17, 18].

Heat-induced instability occurs due to the degradation of the halide perovskite itself or the degradation of any other constituent layer of the device, including the transporting layers and the electrode. The halide perovskite MAPbI₃ is prone to decompose into methyl iodide (CH₃I), ammonia (NH₃) and lead iodide (PbI₂) under elevated temperatures applied as long-term treatment [19]. This decomposition process is proved to be a surface-initiated layer-by-layer degradation [20]. The low formation energy of MAPbI₃ of 0.11–0.14 eV, close to the thermal energy of 85 °C, is an inherent cause [13]. High temperatures (e.g. > 70 °C) can also greatly accelerate the inter-diffusion of elements in component layers, e.g. iodine from perovskite to transporting layers, or electrode diffusion to the perovskite layer, leading to the severe degradation of the transporting properties of PSCs [21].

Briefly, there are four modes of bias voltage applied on solar cell modules in the field, including maximum power point (MPP) voltage (normal operation), reverse bias voltage (shaded modules), open circuit (disconnected) and short circuit (shunting). The degradation caused by bias voltage is mainly ascribed to the migration of ions present in the halide perovskite (e.g. Γ and MA^+) [22]. As a result, the accumulated charges at the interfaces lead to the formation of radicals and the observation of hysteresis in IV curves. Moreover, the induced external ion diffusion into the transporting layers and metal contacts provoke the formation of carrier traps and detrimental reactions [22–24]. In particular, it is reported that a reverse bias voltage of merely -1 to -4 V can cause the breakdown of perovskite solar cells, while the breakdown value for silicon solar cells is around -15 V [24]. The mechanism is considered to be the shunt path formation or the electrochemical reaction of defects in the contacts [24].

2.2. Combination of stress factors

In general, the combination of stress factors results in more severe degradation to PSCs due to the synergetic effects and interactions between different failure mechanisms [25]. For instance, elevated temperature accelerates all involved chemical reactions, ion migration and diffusion due to the increased thermal energy of atoms. Light excites electrons and holes in the perovskite thin films, leading to further reactions with neighbouring materials and the interaction with point defects.

Light coupled with humidity can further catalyse perovskite to form I₂, resulting in irreversible reactions and trapped charge-driven degradation [26, 27]. Compared to conditions under a single stress factor, MA⁺ based halide perovskites decompose faster in the presence of light and oxygen or under bias voltage and oxygen. This is ascribed to the reaction of MA^+ with the superoxide O_2^- , which originates from the reaction of photo-excited electrons in perovskite and molecular oxygen [28, 29]. Light coupled with oxygen and bias voltage is likely to cause accelerated irreversible degradation starting from defective grain boundaries [30]. When PSCs are aged under one sun with MPP tracking, at room temperature and under inert gas, stability depends mainly on ion and defect migration taking place at the constituent materials and interfaces [31, 32]. Interestingly, PSCs have already passed the damp-heat standard stability test for 1000 h by several groups [33, 34], but it is still challenging to overcome tests under continuous light irradiation at elevated temperatures. Stable PSCs under 85 °C and one sun illumination for 1000 h have not been reported yet, though promising results for 500 h have been demonstrated [35, 36]. Moreover, under constant light and heat, PSCs perform worse if analysed under open circuit voltage conditions than under the MPP tracking mode, indicating that ion migration and charge accumulation are critical for the stability of PSCs [37, 38]. Therefore, the long-term stability of PSCs analysed under the combination of light, elevated temperature and bias voltage is significantly important and can be considered as a prerequisite towards commercialization. An important challenge originates from I_2 -induced degradation [38, 39], thermal decomposition, ion migration, interface deterioration and their coupling with defects, leading to accelerated degradation processes. Particularly, the generation of I_2 vapour, which originates from either the oxidation of I^- by photo-generated holes [40] or decomposition of the halide perovskite through the presence of defects in the bulk, is a self-catalysed reaction, and thus urgent innovative solutions are necessary to address this problem [38].

3. Current strategies for stability improvement

Several strategies have been proposed and applied to improve the long-term stability of PSCs. Some of them aim to prevent the permeation of humidity and oxygen and to block the out-diffusion of ions and volatile species from the perovskite layer. This has been achieved by interface engineering or by the application of stable transporting layers, electrodes and interlayers [41–45]. Further strategies focus on the perovskite layer itself, which can be compositional and additive engineered, to enhance its robustness under stress factors [32, 36, 46, 47]. This is realized by the replacement of cations or anions shaping more stable perovskite structures or by the decrease and passivation of detrimental defects. Based on different published strategies, ongoing representative routes are discussed in the following sections.

3.1. Triple-layer carbon based mesoscopic PSCs

The carbon-based mesoscopic PSCs fabricated by screen printing have demonstrated the best long-term operational stability of over 10 000 h [2], representing a promising low-cost technology route for PSC marketization. The devices feature a hole-conductor-free printable triple-layer architecture made of mesoporous TiO_2 (mp- TiO_2) as the ETL, mesoporous ZrO_2 (mp- ZrO_2) as the scaffold and mesoporous carbon as the back electrode [48]. Perovskite applied here is mainly MAPbI₃. The superior stability stems from the special mesoporous structure of thermally-stable inorganic metal oxide, the hydrophobic carbon electrode and the 2D/3D mixed composition realized by the addition of 5-ammonium valeric acid iodide (5-AVAI) [2, 48]. Specifically, the mp-TiO₂ protects the perovskite from external degradation factors like humidity, vapours and impurity diffusions [31]; the mp-ZrO₂ prevents the electrons from flowing into back contact recombining with the holes in the perovskite/carbon interface, as well as acting as another protective layer surrounding perovskite [48]; the thick carbon back electrode (around 10 μ m) effectively serves as a water-retaining layer avoiding the moisture attack on perovskite [48]; the additive 5-AVAI can form a template for crystal nucleation and growth of perovskite and can cross-link the perovskite crystals facilitating defects passivation and carrier transport, as well as increasing the loading of perovskite inside the mp-TiO₂ [48, 49]. The advantage of the triple-layer carbon based mesoscopic PSCs is their lost-cost process avoiding the use of hole transporting layer and noble metals like Au and Ag, together with the exceptional long-term stability. Nevertheless, the efficiency of this type of PSC still falls far behind the hybrid PSCs (<17%) [50], which is mainly due to the poor contact at the perovskite/carbon interface, and the relatively high resistance of bulk carbon [51]. A good interface contact is depending on the porosity and wettability of carbon, which enables the diffusion and loading of perovskite into the porous structure. Various strategies have been performed to mitigate these issues, by using malleable ultrathin graphite [52], pore forming agent [53], post-treatments [54], and interlayers [55].

3.2. Inorganic PSCs

An emerging approach to stable PSCs is the application of all inorganic perovskite compositions, which avoids the use of unstable organic cations. Recently, CsPbI₃ PSCs have achieved over 18% certified efficiency, marking a record in all types of inorganic halide PSCs [50]. The devices have demonstrated 500 h under continuous illumination at room temperature [56], which is a huge step forward in terms of stability. Although inorganic halide PSCs are intrinsically and thermodynamically more stable than organic-inorganic counterparts, the phase stability at operating temperature still requires further investigation, e.g. through additive and defect engineering [57–59].

3.3. 2D PSCs

Another concept is the use of bulkier hydrophobic organic cations to replace the hygroscopic MA and FA organic ions. Due to tolerance factors, the use of a large bulkier organic cation results in the transformation of the 3D perovskites into the 2D layered perovskite structure, which is environmentally more stable [60, 61]. Although more efforts to enhance device efficiency are required [62], the adoption of 2D/3D hybrid structures or 2D interlayers is promising routes to explore the advantages of both types of perovskites. For instance, hybrid 2D/3D PSCs have achieved the longest reported operational stability, currently at 10 000 h (around 1 year), by introducing the 5-AVAI molecule within the halide perovskite absorber [2]. The application of the n-butylammonium additive resulted in 2D/3D PSCs above 17% efficiency, showing 1680 h stability under continuous illumination and open circuit voltage conditions [63]. In addition, the employment of an ultrahydrophobic 2D perovskite interlayer of (FEA)₂PbI₄ (FEA = henylethylammonium) onto the 3D perovskites surface, led to a 2D/3D PSCs. This type of PSCs with over 22% can retain 90% of the initial efficiency during continuous operation for 1000 h in humid air under simulated sunlight (unsealed device) [64].

3.4. 3D hybrid organic-inorganic PSCs

The 3D hybrid organic-inorganic PSCs hold the world record in efficiency among all types of PSCs and thin film photovoltaic technologies [1]. If the stability issues can be solved, it will definitely become the next commercialized photovoltaic technology. Enormous efforts have been made to improve the stability as well as the efficiency of hybrid PSCs. Recently, hybrid PSCs with an efficiency higher than 20% have demonstrated excellent stability of over 1000 h under one sun at temperatures above 60 °C. This is achieved by using ionic liquid additives [46], or via interface engineering with wide band gap PbSO₄ salts [43] or chlorinated graphene oxide [42], or through the replacement of unstable N2,N2,N2',N2',N7,N7,N7'-octakis(4-methoxyphenyl)-9,9'-spirobi[9 H-fluorene]-2,2',7,7'-tetramine (Spiro-OMeTAD) hole transporting layer (HTL) by the inorganic CuSCN [44]. The estimated T₈₀ (time at a performance loss of 20%) of high efficiency hybrid PSCs at 70 °C–75 °C is less than 6000 h [46], indicating a huge improvement, although there is still considerable research required before realizing practical

applications.

Table 1 summarises the most stable PSCs reported in the literature to date following real stability test conditions, meaning analyses under outdoor conditions or under indoor conditions exposed to a combination of light, heat and bias voltage. We have included only those PSC devices able to stand for more than 500 h under one sun at temperatures above 50 °C and retaining more than 80% of their initial efficiency.

4. Multi-component engineering for enhanced stability

In general, halide perovskite with a Cs-containing composition has been proven to be more stable than the sole use of MAPbI₃ due to the intrinsically stable structure and enhanced chemical bonds [65–67]. Moreover, MA-free perovskite has higher thermal stability [31]. The 2D perovskite degrades much more slowly under humidity in comparison with 3D perovskite due to its superior hydrophobic nature [61, 64]. Inorganic transporting layers are thermally more stable than organic counterparts. For HTLs working at temperatures > 60 °C, the poly-[bis (4-phenyl) (2,4,6-trimethylphenyl) amine] (PTAA) outperforms the most commonly used HTL Spiro-OMeTAD in stability, due to its lower permeation of the Au metal electrode [36, 37].

Table 1 shows that the long-term operational stability of PSCs can be achieved regardless of the type of device: inverted or normal configuration, planar or mesoporous structure, or carbon-based architecture. The most important aspect is the robustness of each of the constituent materials of the device: the halide perovskite, the transporting layers (HTLs and ETLs), the electrodes and their interfaces. It can also be observed that when the stress temperature is below 70 °C, the application of a single strategy, such as the modification of one component of the device, can lead to stable PSCs. Some of these strategies include the use of inorganic transporting layers or interlayers [42, 68], the engineering of perovskite composition [63, 69], or the encapsulation with highly hydrophobic photocurable fluoropolymers [70]. Above 70 °C,

rsis.	
. <u>s</u>	
Ĥ	
13	
aı	
Þ.	
E.	
pi	
ta	
S	
eı	
τĤ	
a	
5	
u:	
ъ	
Щ	
еĿ	
-	
Ξ	
- E	
-=	
-Ħ	
ě	
1	
ō	
%	
ö	
8	
ц	
Ja	
Ţ	
re	
ō	
Ξ	
50	
цц,	
.ц	
ar	
Ъ,	
311	
ñ	
Υ,	
Ű	
0	
0	
ŝ	
ve	
õ	
đ	
s,	
ĕ	
n	
ati	
Ľ.	
ъ	
ä	
eı	
ц. Ц	
at	
Ч	
20	
50	
Ē	
la.	
ťh	
e 1	
Е	
ŭ	
ч.	
or	
-	
ŋ, f	
sun, f	
l sun, f	
f 1 sun, f	
of 1 sun, f	
n of 1 sun, f	
ion of 1 sun, f	
ttion of 1 sun, f	
nation of 1 sun, f	
rination of 1 sun, f	
umination of 1 sun, f	
llumination of 1 sun, f	
illumination of 1 sun, f	
us illumination of 1 sun, f	
ous illumination of 1 sun, f	
uous illumination of 1 sun, f	
inuous illumination of 1 sun, f	
ntinuous illumination of 1 sun, f	
ontinuous illumination of 1 sun, f	
r continuous illumination of 1 sun, f	
ler continuous illumination of 1 sun, f	
ıder continuous illumination of 1 sun, f	
under continuous illumination of 1 sun, f	
1 under continuous illumination of 1 sun, f	
ed under continuous illumination of 1 sun, f	
yzed under continuous illumination of 1 sun, f	
alyzed under continuous illumination of 1 sun, f	
malyzed under continuous illumination of 1 sun, f	
, analyzed under continuous illumination of 1 sun, f	
re, analyzed under continuous illumination of 1 sun, f	
ure, analyzed under continuous illumination of 1 sun, f	
ature, analyzed under continuous illumination of 1 sun, f	
erature, analyzed under continuous illumination of 1 sun, f	
literature, analyzed under continuous illumination of 1 sun, f	
e literature, analyzed under continuous illumination of 1 sun, f	
the literature, analyzed under continuous illumination of 1 sun, f	
1 the literature, analyzed under continuous illumination of 1 sun, f	
in the literature, analyzed under continuous illumination of 1 sun, f	
$\dot{c}d$ in the literature, analyzed under continuous illumination of 1 sun, f	
ted in the literature, analyzed under continuous illumination of 1 sun, f	
orted in the literature, analyzed under continuous illumination of 1 sun, f	
ported in the literature, analyzed under continuous illumination of 1 sun, f	
reported in the literature, analyzed under continuous illumination of 1 sun, f	
$\mathfrak{I}_{\mathfrak{I}}$ reported in the literature, analyzed under continuous illumination of 1 sun, f	
5Cs reported in the literature, analyzed under continuous illumination of 1 sun, f	
PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
e PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
ble PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
table PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
ly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
thly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
lighly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
Highly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
1. Highly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
e 1. Highly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
ble 1. Highly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	
able 1. Highly stable PSCs reported in the literature, analyzed under continuous illumination of 1 sun, f	

inc sri nrini c			Stability				
contact	Testing time (h) Initial		Str	ess factor			
skite Back ct	eff. (% Relative eff. retaining (%)	Light source	(°C)	Bias voltage	RH (%) and Oxygen	Strategy employed to enhance stability	Ref
EA0.83 MA0.17)0.95 Pb(I0.9Br0.1)3 ABF4 PCBM/ Tr(Cr>O3)/Au	1885 ~18 ~85	Xenon Lamp with UV	70–75	Open circuit	Sealed, 40–50 air	 (1)Interlayer: Cr(Cr₂O₃); (2)Additive: Ionic liquid BMIMBF₄; (3)Composition engineering:Cs containing. 	46
Cso.05FA0.81MA0.14 Br0.45 /C60/BCP/Cu	1200 19.44 96.8	Plasma lamp with UV	65	MPP	Sealed $\sim 60 \pm 10$ air	 (1)Interface engineering: PbSO₄ layer; (2)Encapsulation: CYTOP fluoropolymers; (3)Composition engineering Cs containing. 	43
28.05 MA0.95 CBM/BCP/ Ag/Al2O3	500 16.5 86.7	Xenon lamp No UV	85	MPP	Unsealed 20–60 air	 (1)Outer layer passivation: Ål₂O₃; (2)Interlayer: AZO; (3)Composition engineering: Cs contain- ing. 	35
([Li)O JI M Ox/Ag	1000 - 90	Solar simulator No UV	45-50	Short circuit	Sealed air	Li)Inorganic HTL and ETL: Metal oxide NiMg(Li)O and Ti(Nb)Ox	68

		Ref	ffusion 34 ride		87		44	44	44 iin-	44 iin- yer; 47	44 uin- tyer; 47	44 iin- yeu; 47 iining
		Strategy employed to enhance stability	(1) Interface engineering: 2D iodide di blocking barrier: graphitic carbon nit	(g-C3IN4)	(1) Interface engineering: Trimethyl- olnronane triacrylate (TMTA)	(2) Additive: TMTA	(2) Additive: TMTA (1)Inorganic H7L: CuSCN;	 (2) Additive: TMTA (1) Inorganic HTL: CuSCN; (2) Interlayer: r-GO; (3) Commonition antineering Contraction 	 (2) Additive: TMTA (1)Inorganic HTL: CuSCN; (2)Interlayer: r-GO; (3)Composition engineering: Cs containg. 	 (2) Additive: TMTA (1)Inorganic HTL: CuSCN; (2)Interlayer: r-GO; (3)Composition engineering: Cs containg. (1)Interface engineering: SN surface la 	 (2) Additive: TMTA (1)Inorganic HTL: CuSCN; (2)Interlayer: r-GO; (3)Composition engineering: Cs containg. (1)Interface engineering: SN surface la (2)Additive: SN molecular modulator; 	 (2) Additive: TMTA (1)Inorganic HTL: CuSCN; (2)Interlayer: r-GO; (3)Composition engineering: Cs containg. (1)Interface engineering: SN surface la (2)Additive: SN molecular modulator; (3)Composition engineering: Cs conta
		RH (%) and Oxygen	Sealed air		Unsealed Glove box		Unsealed	Unsealed N2	Unsealed N2	Unsealed N2 Unsealed	Unsealed N ₂ Unsealed Ar	Unsealed N ₂ Unsealed Ar
	ss factor	Bias voltage	MPP		MPP		MPP	MPP	MPP	MPP MPP	MPP	MPP MPP
Stability	Stree	(C) (D) (D) (D) (D) (D) (D) (D) (D) (D) (D	60		60		60	60	60	60 55-60	60 55-60	60 5560
		Light source	Solar simulator No UV		White LED No UV		White LED	White LED No UV	White LED No UV	White LED No UV Whtie	White LED No UV Whtie LED	White LED No UV Whtie LED No UV
	Testing time (h) Initial	eff. (% Relative eff. :etaining (%)	1000 ~15	1	1000 -	30	30 1000	30 1000 ~20	30 1000 25	30 ~20 ∋5 1000	30 1000 →20 95 1000 20.9	30 1000 ∋5 1000 20.9 88.1
se structure	ont contact	rovskite Back ntact	O 0.85 MA0.15Pb(I0.85	0.15)3 PCBM/BCP/Ag	CT-N 4.MA)Ph(I.Br)3	<pre>`BM/C60/TPBi/Cu</pre>	,BM/C ₆₀ /TPBi/Cu ΓiΟ ₂ /)BM/C60/TPBi/Cu fiO ₂ / 2-TiO2 5 FA MA)Dh/T R-1.	,BM/C ₆₀ /TPBi/Cu TiO ₂ / >-TiO ₂ s,FA,MA)Pb(I,Br) ₃ SCN/r-GO/Au	BM/C ₆₀ /TPBi/Cu IIO ₂ / -TiO ₂ s,FA,MA)Pb(I,Br) ₃ s,SCN/r-GO/Au IIO ₂ /	BM/C ₆₀ /TPBi/Cu IiO ₂ / -TiO ₂ s,FA,MA)Pb(I,Br) ₃ SCN/r-GO/Au IiO ₂ / P-TiO ₂	BM/C ₆₀ /TPBi/Cu TiO ₂ / -TiO ₂ s,FA,MA)Pb(L,Br) ₃ s,FA,MA)Pb(L,Br) ₃ s,FA,MA)Pb(J,Br) ₃ s,FA,MA)Pl3; SN*
Devic	Frc	Pei Configuration con	Ni FA		P3 (F/	PC	PC [-]	PC II C	27 2.5 1 1 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	PC 6.1 0.0 0.0 0.0 0.0 0.0 0.0 0.0 0.0 0.0 0	PC c-1 (C. Vormal c-1 mesoporous mj	PC c-1 (C. C. Normal nesoporous FA

Device structure			Stability				
Front contact	Testing time (h) Ini- tial		St	ress factor			
Perovskite Back on contact	eff. (% Relative eff. retaining (%)	Light source	T (°C)	Bias voltage	RH (%) and Oxygen	Strategy employed to enhance stability	Ref
c-TiO ₂ / mp-TiO ₂ (Rb,Cs,FA,MA)Pb(I,Br) PTAAAI	500 ~14 3 95	Whtie LED No UV	85	MPP	Unsealed N ₂	(1)HTL: PTAA; (2)Composition engineering: Rb, Cs containing.	36
c-TiO ₂ / mp-TiO ₂ / (FAI) _{0.81} (PbI ₂) _{0.85} (MABr) _{0.15} (PbBr ₂) _{0.15} : BrPh-ThR* ibis-PCBM Spiro/Au	1500 - 90	Whtie LED No UV	55	MPP	Unsealed N2	(1)Additive engineering: Lewis base and acid: BrPh-ThR and bis-PCBM	88
c-TiO ₂ / mp-TiO ₂ / (FA,MA)Pb(I,Br) ₃ Spiro/Au/fluoronolymer	2208 - 95	Outdoor, In Turin, Sun	-3-27	Open circuit	Unsealed, 27 Rainy days	(1)Encapsulation: photocurable fluoropolymers protective layer	70
c-TiO ₂ / mp-TiO ₂ MAPbI ₃ PTAA/Au	500-85	White LED, No UV	95	MPP	Unsealed, N2	(1)HTL: PTAA	37
c-TiO ₂ / mp-TiO ₂ MA _{0.75} GUA _{0.25} PbI ₃ Spiro/Au	1100 17.14 80	Indoor, one sun	60	MPP	Unsealed, Ar	(1)Composition engineering: Guanidinium mixi	1 g 69

				Table 1. (Conti	nued)			
	Jevice structure			Stability				
	Front contact	Testing time (h) Initial		Str	ess factor			
Configuration	Perovskite Back contact	eff. (% Relative eff. retaining (%)	Light source	T (°C)	Bias voltage	RH (%) and Oxygen	 Strategy employed to enhance stability 	Ref
Normal Plana	- SnO ₂ /PCBM BA _{0.09} (FA _{0.83} Cs _{0.17}) _{0.91}	1680 -	xenon lamp With UV	50-60	Open circuit	Sealed, 40–50,	(1)Composition engineering: 2D/3D mixture and Cs containing MA-free.	63
	$Pb(I_{0.6}Br_{0.4})_3 Spiro/Au$	80		0		air 2 1 1 -		:
	SnO ₂ (FA MANDLI	1000	AM 1.5G,	60	MPP	Sealed, air	(1)Interface engineering: Cl-GO interlayer;	42
	(FA,MA)P013 Cl-GO/PTAA/Au	21.4 90	one sun				(Z)H1L: F1AA.	
Carbon	c-TiO ₂ /	1000	Solar simulator,	50	MPP	Unsealed,	(1)Mesoporous metal oxide;	89
based	$mp-TiO_2/$	8	With UV			54,	(2)Hydrophobic carbon layer;	
	mp-ZrO2 MAPbI3: 5-AVAI	100				air	(3)Additive:5-AVAI leading to 2D/3D composi- tion.	
	mp-carbon							
	4	720	Outdoor,	30	MPP	Sealed module		89
		8	In Wuhan,			80,		
		100	Sun			air		
		10 000	Solar simulator	55	Short circuit	Sealed module		2
		$\sim\!10$	No UV			air		
		100						



different strategies are combined to achieve exceptional long-term stability [35, 36, 46]. In particular, when aging for over 1000 h at 70 $^{\circ}$ C–75 $^{\circ}$ C under continuous light irradiation and open circuit conditions, the simultaneous engineering of perovskite composition, additive content and interlayers is employed [46]. An increase in testing temperature of only 10 $^{\circ}$ C could result in the acceleration of PSC degradation by a factor of two

[4, 46], suggesting that more sophisticated methods must be employed to overcome instability. We should highlight that the current most stable PSCs (the Carbon-based PSCs) also involve the strategy of multi-component engineering. It adopts a 2D/3D perovskite composition, uses a mixture of mesoporous oxides (as transport and scaffold layers) and applies hydrophobic carbon electrodes simultaneously [2, 48]. We note that in table 1, MAPbI₃-based PSCs with a simple strategy applying PTAA HTL retained 85% of their initial efficiency at 95 °C under MPP tracking for 500 h [37]; however, under similar aging conditions at 65 °C, PSCs based on CsFAMA triple-cation composition retained 97% performance versus 90% remaining efficiency of MAPbI₃-based PSCs [37]. This again demonstrates that the Cs-containing composition is superior to pure MAPbI₃ counterparts, and even more enhanced stability can be expected if multi-component engineering is applied.

In summary, composition and additive engineering are necessary to improve the inherent photo-thermal stability of perovskite at 85 °C under one sun, while other strategies, including the application of inorganic transporting layers and barrier interlayers, stable electrodes and interface engineering, are crucial to prevent humidity and oxygen incursion into the device. Many of these strategies play an important role in passivating defects at the grain boundaries and interfaces, mitigating the migration of ions. Equally important is the engineering of PSCs considering an economic perspective. Expensive transporting materials, such as PTAA and [6]-phenyl-C⁶¹ -butyric acid methyl ester (PCBM), may not be favourable for large-scale production. Because the ultimate aim is to achieve a levelized cost of electricity (LCOE) value comparable with other energy resources for future commercialization, low cost, high efficiency and long-term stability should be considered comprehensively [71].

Therefore, the long-term operational stability of PSCs requires a rational design of each of the constituent materials of the device; however, most importantly, a robust device can only be developed through the simultaneous engineering of the different constituent materials and interfaces of the PSCs.

Table 2 summarises (a) the main stress factors and (b) the failure modes that affect the stability of PSCs. We have also shown (c) the methods currently applied to engineer the different components of a PSC to enhance its efficiency and stability.

Thus, in an attempt to achieve highly efficient PSCs with long-term operational stability, we highlight the strategy of multi-component engineering based on a device configuration shown in figure 1. In addition to the common constituents of a PSC, this configuration also includes an encapsulation layer and the introduction of three interlayers, two of which are placed between perovskite and transporting layers and a

		(a) Ma	in stress factor	S	(b) Main failure			(c) Strategies for sta	ability improvemen	ıt	
Device			Bias	Humidity	modes	Halide I	perovskite	Transport			
structure	Light	Heat	voltage	and Oxygen		Composition	Additive	Layer	Interfaces	Electrodes	Encapsulation
Front Contacts	2	2	2	1	UV instability, interface photochemical reaction, detrimental ion diffusion to perovskite			Apply thermally stable materials: Metal oxides (e.g. NiO _x , SnO ₂ , CuGaO ₂), complex oxides (e.g. LaxBaySnO ₃ , PbZrTiO ₃), Metal sulfides (e.g. MnS), CuSCN, or their combinations	Application of UV and thermally stable Interlayers for interface passivation and band alignment optimization. (e.g. CsBr, ZnS, 2D transition metal carbides, nitrides and carbonitrides (MXenes))	1	
ovskite	>	2	2	>	defects and ion migration, thermal decomposition, hydrated reaction, light and oxygen induced superoxide, I ₂ induced degradation under light and heat	Improve photo-thermal stability and environmental stability by applying Cs-containing and MA-free halide perovskites (e.g. (Cs,FA) Pb(I,Br) ₃) (I,Br) ₃)	Enhance photo-thermal stability, passivation of defects and immobilize ion migration by the use of additives (e.g. NaE,Eu ³⁺ , Ionic liquid BMIMBF4, Caffeine, 2D perovskite) (Continued)	1	1	Ι	I

									1		
		(a) Mai	n stress factor	S	(b) Main failure			(c) Strategies for sta	ability improvemen		
Device			Bias	Humidity	modes	Halide J	perovskite	Transport			
structure	Light	Heat	voltage	and Oxygen		Composition	Additive	Layer	Interfaces	Electrodes	Encapsulation
Back		7	2	>	Heat induced			Employ Metal	Use of thermally	Use of Chem-	Use of thermal,
Contacts					structure			oxides (e.g.	stable	ical stable con-	photo and air-
					transformation,			NiO_x , SnO_2 ,	Interlayers to	ductors (e.g.	stable encapsu-
					electric bias			$Ti(Nb)O_x$,	block ion	Carbon, TCO)	lation materials.
					induced ion			$CuGaO_2$),	migration,		(e.g.
					diffusion from			complex oxides	humidity and		polyolefin,
					perovskite,			(e.g.	oxygen		fluoropolymers,
					humidity and			$La_X Ba_Y SnO_3$,	incursion.		epoxy resin,
					oxygen			PbZrTiO ₃),	(e.g. PbSO ₄ ,		(ER)-based
					penetration			Metal sulfides	Al ₂ O ₃ , rGO,		polymers)
								(e.g. MnS),	Cl-GO, MnO ₃ ,		
								CuSCN,	Ta-WOx,		
								Or their	Cr/Cr ₂ O ₃ ,2D		
								combinations	$g-C_3N_4$, 2D		
									perovskite)		

11

third between the back electrode and transporting layers. The strategy of multi-component engineering consists of the simultaneous application of several of the strategies described in table 2, section (c), such as:

- Perovskite layer: the use of an MA-free and Cs-containing composition along with additives (e.g. BMIMBF₄, Caffeine, NaF, SN, Eu³⁺, etc) to enhance the PSC photo-thermal stability, to passivate defects, to minimize I₂ vapor generation and to immobilize ion migration of perovskite layer [31, 32, 46, 47, 69, 72, 73].
- Transporting layers: to apply thermal, photo and air-stable materials, such as metal oxides and metal sulphides [17, 18, 68, 74–80].
- Interlayers: to consider the engineering of interfaces applying thermally stable materials, which will minimize ion diffusion, interface recombination and air ingress. For example, the use of PbSO₄, Cl-GO, 2D materials, metal oxides or sulphides, among others [42–44, 46, 64, 81, 82]. In particular, Interlayer 1 is for the minimization of interfacial reactions due to the photocatalytic effect, Interlayer 2 is for blocking ion diffusion from perovskite and Interlayer 3 is for preventing humidity and oxygen incursion and electrode diffusion. All the interlayers can also play an important role in interfacial defect passivation and band alignment optimization.
- Electrodes: to select chemical stable conductors (e.g. carbon, transparent conductive oxides (TCO)) [33, 48, 83].
- Encapsulation: to apply water and oxygen resistant materials, e.g. polyolefin, fluoropolymers or 2D materials [70, 84–86].

5. Outlook

Given the high efficiency of hybrid organic-inorganic PSCs, the research focus should be shifted to long-term stability under more realistic operational conditions, and specifically under a combination of light, heat and bias voltage. More innovative solutions and strategies are still urgently needed to tackle the I₂-induced degradation in the bulk and to improve the long-term photo-thermal stability at 85 °C and under one sun. To achieve a lifetime of 25 years for PSCs, multi-component engineering is an option worthy of attention. The application of this multi-component engineering strategy may encourage researchers to develop PSC devices with enhanced stability under more stressful but practical operational conditions. This can guarantee a competitive technology that can realize commercialization more rapidly.

Acknowledgments

M L C and H X acknowledges the support from Spanish MINECO for the grant GraPErOs (ENE2016-79282-C5-2-R), the OrgEnergy Excellence Network CTQ2016-81911- REDT, the Agència de Gestiód'Ajuts Universitaris i de Recerca (AGAUR) for the support to the consolidated Catalonia research group 2017 SGR 329 and the Xarxa de Referència en Materials Avançats per a l'Energia (Xarmae). ICN2 is supported by the Severo Ochoa program from Spanish MINECO (Grant Nos. SEV-2017-0706) and is funded by the CERCA Programme/Generalitat de Catalunya.

ORCID iDs

Haibing Xie https://orcid.org/0000-0002-5070-2882 Monica Lira-Cantu https://orcid.org/0000-0002-3393-7436

References

- [1] NREL 2020 Best Research-Cell Efficiency Chart https://nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.20200406.pdf
- [2] Grancini G et al 2017 Nat. Commun. 8 15684
- [3] Holzhey P and Saliba M 2018 J. Mater. Chem. A 6 21794-808
- [4] Snaith H J and Hacke P 2018 Nat. Energy 3 459-65
- [5] Saliba M, Stolterfoht M, Wolff C M, Neher D and Abate A 2018 Joule 2 1019–24
- [6] Khenkin M V et al 2019 Nat. Energy 5 35-49
- [7] Reese M O et al 2011 Sol. Energy Mater. Sol. Cells 95 1253-67
- [8] Boyd C C, Cheacharoen R, Leijtens T and McGehee M D 2019 Chem. Rev. 119 3418–51
- [9] Wang R, Mujahid M, Duan Y, Wang Z-K, Xue J and Yang Y 2019 Adv. Funct. Mater. 0 1808843
- [10] Ono L K, Qi Y and Liu S 2018 Joule 2 1961–90
- [11] Shirayama M, Kato M, Miyadera T, Sugita T, Fujiseki T, Hara S, Kadowaki H, Murata D, Chikamatsu M and Fujiwara H 2016 J. Phys. D: Appl. Phys. 119 115501
- [12] Christians J A, Miranda Herrera P A and Kamat P V 2015 J. Am. Chem. Soc. 137 1530-8
- [13] Conings B et al 2015 Adv. Energy Mater. 5 1500477
- [14] Leijtens T, Eperon G E, Pathak S, Abate A, Lee M M and Snaith H J 2013 Nat. Commun. 4 2885

- [15] Ito S, Tanaka S, Manabe K and Nishino H 2014 J. Phys. Chem. C 118 16995-7000
- [16] Zhang P, Wu J, Wang Y, Sarvari H, Liu D, Chen Z D and Li S 2017 J. Mater. Chem. A 5 17368–78
- [17] Pérez-Tomas A et al 2019 Sustainable Energy Fuels 3 382-9
- [18] Shin S S, Yeom E J, Yang W S, Hur S, Kim M G, Im J, Seo J, Noh J H and Seok S I 2017 Science 356 167–71
- [19] Juarez-Perez E J, Hawash Z, Raga S R, Ono L K and Qi Y 2016 Energy Environ. Sci. 9 3406-10
- [20] Fan Z et al 2017 Joule 1 548–62
- [21] Domanski K, Correa-Baena J-P, Mine N, Nazeeruddin M K, Abate A, Saliba M, Tress W, Hagfeldt A and Grätzel M 2016 ACS Nano 10 6306–14
- [22] Khenkin M V, K. M A, Katz E A and Visoly-Fisher I 2019 Energy Environ. Sci. 12 550-8
- [23] Bae S, Kim S, Lee S-W, Cho K J, Park S, Lee S, Kang Y, Lee H-S and Kim D 2016 J. Phys. Chem. Lett. 7 3091-6
- [24] Bowring A R, Bertoluzzi L, O'Regan B C and McGehee M D 2018 Adv. Energy Mater. 8 1702365
- [25] Domanski K, Alharbi E A, Hagfeldt A, Grätzel M and Tress W 2018 Nat. Energy 3 61-67
- [26] Niu G, Li W, Meng F, Wang L, Dong H and Qiu Y 2014 J. Mater. Chem. A 2 705–10
- [27] Ahn N, Kwak K, Jang M S, Yoon H, Lee B Y, Lee J-K, Pikhitsa P V, Byun J and Choi M 2016 Nat. Commun. 7 13422
- [28] Bryant D, Aristidou N, Pont S, Sanchez-Molina I, Chotchunangatchaval T, Wheeler S, Durrant J R and Haque S A 2016 Energy Environ. Sci. 9 1655–60
- [29] Aristidou N, Sanchez-Molina I, Chotchuangchutchaval T, Brown M, Martinez L, Rath T and Haque S A 2015 Angew. Chem., Int. Ed. 54 8208–12
- [30] Gomez A, Sanchez S, Campoy-Quiles M and Abate A 2018 Nano Energy 45 94–100
- [31] Turren-Cruz S-H, Hagfeldt A and Saliba M 2018 Science 362 449-53
- [32] Wang L et al 2019 Science 363 265–70
- [33] Bush K A et al 2017 Nat. Energy 2 17009
- [34] Bi E et al 2019 Joule 3 2748–60
- [35] Seo S, Jeong S, Bae C, Park N-G and Shin H 2018 Adv. Mater. 30 1801010
- [36] Saliba M et al 2016 Science 354 206-9
- [37] Holzhey P, Yadav P, Turren-Cruz S-H, Grätzel M, Hagfeldt A and Saliba M 2018 Mater. Today 29 10-19
- [38] Fu F et al 2019 Energy Environ. Sci. 12 3074-88
- [39] Wang S, Jiang Y, Juarez-Perez Emilio J, Ono Luis K and Qi Y 2016 Nat. Energy 2 16195
- [40] Kim G Y, Senocrate A, Yang T-Y, Gregori G, Grätzel M and Maier J 2018 Nat. Mater. 17 445–9
- [41] Jung E H, Jeon N J, Park E Y, Moon C S, Shin T J, Yang T-Y, Noh J H and Seo J 2019 Nature 567 511–15
- [42] Wang Y, Wu T, Barbaud J, Kong W, Cui D, Chen H, Yang X and Han L 2019 Science 365 687–91
- [43] Yang S et al 2019 Science 365 473-8
- [44] Arora N, Dar M I, Hinderhofer A, Pellet N, Schreiber F, Zakeeruddin S M and Grätzel M 2017 Science 358 768–71
- [45] Lira-Cantú M 2017 Nat. Energy 2 17115
- [46] Bai S et al 2019 Nature 571 245-50
- [47] Bi D et al 2018 Nat. Commun. 9 4482
- [48] Mei A et al 2014 Science 345 295-8
- [49] Li X, Ibrahim Dar M, Yi C, Luo J, Tschumi M, Zakeeruddin S M, Nazeeruddin M K, Han H and Grätzel M 2015 Nat. Chem. 7 703–11
- [50] Rong Y, Hou X, Hu Y, Mei A, Liu L, Wang P and Han H 2017 Nat. Commun. 8 14555
- [51] Fagiolari L and Bella F 2019 Energy Environ. Sci. 12 3437–72
- [52] Duan M, Rong Y, Mei A, Hu Y, Sheng Y, Guan Y and Han H 2017 Carbon 120 71–76
- [53] Tao H, Li Y, Zhang C, Wang K, Wang J, Tan B, Han L and Tao J 2018 Solid State Commun. 271 71–75
- [54] Hashmi S G, Martineau D, Dar M I, Myllymäki T T T, Sarikka T, Ulla V, Zakeeruddin S M and Grätzel M 2017 J. Mater. Chem. A 5 12060–7
- [55] Cao K, Zuo Z, Cui J, Shen Y, Moehl T, Zakeeruddin S M, Grätzel M and Wang M 2015 Nano Energy 17 171-9
- [56] Wang Y et al 2019 Science 365 591-5
- [57] Zhou Y and Zhao Y 2019 Energy Environ. Sci. 12 1495–511
- [58] Xiang W et al 2019 Joule 3 205-14
- [59] Xiang W and Tress W 2019 Adv. Mater. 0 1902851
- [60] Thrithamarassery Gangadharan D and Ma D 2019 Energy Environ. Sci. 12 2860-89
- [61] Tsai H et al 2016 Nature 536 312
- [62] Luo T, Zhang Y, Xu Z, Niu T, Wen J, Lu J, Jin S, Liu S and Zhao K 2019 Adv. Mater. 0 1903848
- [63] Wang Z, Lin Q, Chmiel F P, Sakai N, Herz L M and Snaith H J 2017 Nat. Energy 2 17135
- [64] Liu Y et al 2019 Sci. Adv. 5 eaaw2543
- [65] Saliba M et al 2016 Energy Environ. Sci. 9 1989–97
- [66] Yi C, Luo J, Meloni S, Boziki A, Ashari-Astani N, Grätzel C, Zakeeruddin S M, Röthlisberger U and Grätzel M 2016 Energy Environ. Sci. 9 656–62
- [67] Lee J-W, Kim D-H, Kim H-S, Seo S-W, Cho S M and Park N-G 2015 Adv. Energy Mater. 5 1501310
- [68] Chen W et al 2015 Science 350 944-8
- [69] Jodlowski A D, Roldán-Carmona C, Grancini G, Salado M, Ralaiarisoa M, Ahmad S, Koch N, Camacho L, de Miguel G and Nazeeruddin M K 2017 Nat. Energy 2 972–9
- [70] Bella F, Griffini G, Correa-Baena J-P, Saracco G, Grätzel M, Hagfeldt A, Turri S and Gerbaldi C 2016 Science 354 203-6
- [71] Meng L, You J and Yang Y 2018 Nat. Commun. 9 5265
- [72] Li N et al 2019 Nat. Energy 4 408-15
- [73] Wang R et al 2019 Joule 3 1464–77
- [74] You J et al 2015 Nat. Nanotechnol. 11 75
- [75] Christians J A, Schulz P, Tinkham J S, Schloemer T H, Harvey S P, Tremolet de Villers B J, Sellinger A, Berry J J and Luther J M 2018 Nat. Energy 3 68–74
- [76] Zhang H, Wang H, Chen W and Jen A K-Y 2017 Adv. Mater. 29 1604984
- [77] Chen R et al 2019 J. Am. Chem. Soc. 141 541-7
- [78] Li X, Yang J, Jiang Q, Lai H, Li S, Tan Y, Chen Y and Li S 2019 J. Mater. Chem. A 7 7065-73
- [79] Roose B, Wang Q and Abate A 2019 Adv. Energy Mater. 9 1803140

- [80] Pérez-Tomás A, Mingorance A, Tanenbaum D and Lira-Cantú M 2018 The Future of Semiconductor Oxides in Next-Generation Solar Cells, ed M Lira-Cantú (Amsterdam: Elsevier) pp 267–356
- [81] Li W, Zhang W, Van Reenen S, Sutton R J, Fan J, Haghighirad A A, Johnston M B, Wang L and Snaith H J 2016 Energy Environ. Sci. 9 490–8
- [82] Hou Y et al 2017 Science 358 1192-7
- [83] Luo Q et al 2018 Adv. Funct. Mater. 28 1706777
- [84] Cheacharoen R, Boyd C C, Burkhard G F, Leijtens T, Raiford J A, Bush K A, Bent S F and McGehee M D 2018 Sustainable Energy Fuels 2 2398–406
- [85] Fu Z et al 2019 Adv. Funct. Mater. 29 1809129
- [86] Jiang Y, Qiu L, Juarez-Perez E J, Ono L K, Hu Z, Liu Z, Wu Z, Meng L, Wang Q and Qi Y 2019 Nat. Energy 4 585–93
- [87] Li X, Fu S, Liu S, Wu Y, Zhang W, Song W and Fang J 2019 Nano Energy 64 103962
- [88] Zhang F, Bi D, Pellet N, Xiao C, Li Z, Berry J J, Zakeeruddin S M, Zhu K and Grätzel M 2018 Energy Environ. Sci. 11 3480–90
- [89] Hu Y, Si S, Mei A, Rong Y, Liu H, Li X and Han H 2017 Sol. RRL 1 1600019