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Melt solidification in the ceramic system CaO-Al₂O₃-SiO₂

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Abstract. Technique of T-x-y diagram for system CaO-Al₂O₃-SiO₂ computer simulation by the use of kinematical surfaces has been suggested. Original 245 crystallization paths were found.

1. Introduction

The system CaO-Al₂O₃-SiO₂ (C-A-S) has a great practical importance [1-2]. Its T-x-y diagram models [3-5] solve series of applied tasks, but don't allow to present full topological structure of diagram for the obtaining of concentration fields with the unique crystallization schemes. In particular, the immiscibility surfaces and 4 liquidus surfaces (C₃S, C₃S₂, C₃A, CA₆) are absent in Vladimir Danek's model [3, P. 147]: "The immiscibility region near the SiO₂ apex was neglected in the calculation as such behavior is not considered in the thermodynamic model. Furthermore, because of the lack of thermodynamic data, the crystallization of rankinite, tricalcium silicate, tricalcium aluminate, and calcium hexaaluminate was not included in the calculation". In the model, considered in [4], there isn't one binary eutectic in system C-A. As a result, a liquidus surface of $C_{12}A_7$ is internal and, unlike the experimental data [5], doesn't adjacent to the binary system. The model calculated by the thermodynamics software CaTCalc [6] doesn't include three liquidus surfaces (C₃S, C₃S₂, C₃A₂).

The kinematical method is used for T-x-y diagram simulation [7], when a kinematical surface is produced by the motion of forming line along to the directing ones described by the interpolating polynomials. Approximating possibilities of the kinematical simulation were checked by the investigation of saddle surface (as the most complex boundary of phase regions) described by the Sheffe's model [8-9]. When kinematical saddle surface was simulated by 7 points (3 simplex vertexes, 3 points on its sides and 1 – within the simplex), a standard deviation with the Sheffe's saddle (in 66 sites of concentration triangle with the step 0,1) was about 2,5 degrees and the most discordance (5-8 degrees) was observed in 17 points. The saddle points differ in the concentration coordinates: polynomial model (0,181; 0.384; 0,435) and kinematical one (0,159; 0.394; 0,447). When the kinematical surface was taken with 3 internal directing curves and simulated by 21 point (3 tops, 3 triads on the simplex' edges and 3 internal triads), the disagreements in temperatures were lowered and the concentration coordinates of saddle points for both models essentially approach each other: 0,194; 0.391; 0,415 – for a kinematical model.

2. Computer model of system CaO-Al₂O₃-SiO₂

The coordinates of 14 binary points (9 eutectics (e) and 5 peritectics (p)), 16 ternary points (6 eutectics (E), 9 quasiperitectics (Q) and the point of four-phase mass regrouping with two allotropies of SiO_2 (U)), 10 binary (R_1 - R_{10}) and 2 ternary compounds (R_{11} and R_{12}) are taken for the computer model

construction [1, 7]. Also the coordinates of inflections on monovariant liquidus curves should be taken into account. The following designation of compounds are used: $R_1 - C_3S$, $R_2 - C_2S$, $R_3 - C_3S_2$, $R_4 - CS$, $R_5 - A_3S_2$, $R_6 - C_3A$, $R_7 - C_{12}A_7$, $R_8 - CA$, $R_9 - CA_2$, $R_{10} - CA_6$, $R_{11} - C_2AS$, $R_{12} - CAS_2$, where R_1 , R_3 , R_6 , R_{10} are ingongruently melting compounds and R_2 , R_4 , R_5 , R_7 , R_8 , R_9 , R_{11} , R_{12} are the congruently melting ones.

The phase diagram involves 16 unruled surfaces (15 liquidus surfaces S and one immiscibility surface i) (table 1, figure 1a), 80 ruled surfaces ($77S^r+3i^r$), 16 horizontal complexes at the invariant points temperatures (H) and 16 vertical planes of triangulation (figure 1b).

Unruled surfaces with more then four points on their contour are constructed by the fragmentation into the more simple parts. In such way the surfaces S_2 , S_4 , S_6 , S_7 , S_{11} , S_{14} , S_{16} are given by two parts, and the surfaces S_9 , S_{12} , S_{13} , S_{15} – by three parts. E.g., surface S_{15} has seven points on contour, the horizontal curve p_3U and the cut mkn remained after intersection with immiscibility surface (figure 2). It includes 3 fragments np_3U , mknUe₄ and SiO₂me₄, simulated by a common directing curve $p_3(n,U)(m,e_4)SiO_2$ and by 2 forming lines n(n,U)U and $m(m,e_4)e_4$. The parts of common directing line are used for the construction of different surface's fragments. So line's parts $p_3(n,U)$, $(n,U)(m,e_4)$ and $(m,e_4)SiO_2$ form the fragments snp_3U , mknUe₄ and SiO_2me_4 accordingly. The cut is given as directing line n(n,k)k(m,k)m with five points on the intermediate surface's fragment. The intermediate point (p_{23},U) should be assigned at temperature $T_{U(p_3)}$ for the horizontal line p_3U generation.

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Figure 1. XY projections of liquidus surfaces and immiscibility cupola (a), all diagram surfaces (b).

16 invariant transformations characterize the considered system. There are nine quasiperitectic transformations (Q), six eutectic ones (E) and one transformation of the mass regrouping with two allotropies of SiO₂ (U): $L(Q_8)+Al_2O_3\rightarrow R_5+R_{12}$, $L(Q_5)+R_9\rightarrow R_8+R_{11}$, $L(Q_7)+Al_2O_3\rightarrow R_{10}+R_{12}$, $L(Q_6)+R_9\rightarrow R_{10}+R_{11}$, $L(Q_1)+CaO\rightarrow R_1+R_6$, $U+SiO^{tr}_2\rightarrow R_5+SiO^{cr}_2$, $L(Q_2)+R_1\rightarrow R_2+R_6$,

 $\begin{array}{ll} L(E_6) \rightarrow R_{10} + R_{11} + R_{12}, & L(Q_4) + R_{11} \rightarrow R_2 + R_8, & L(Q_3) + R_6 \rightarrow R_2 + R_7, & L(E_5) \rightarrow R_5 + R_{12} + SiO_2, \\ L(E_1) \rightarrow R_2 + R_7 + R_8, & L(Q_9) + R_2 \rightarrow R_3 + R_{11}, & L(E_2) \rightarrow R_3 + R_4 + R_{11}, & L(E_3) \rightarrow R_4 + R_{11} + R_{12}, & L(E_4) \rightarrow R_4 + R_{12} + SiO_2. \end{array}$

Phase diagram includes one one-phase region (L), 33 two-phase regions (L₁+L₂, L+CaO, L+R₁, L+R₂, L+R₃, L+R₄, L+R₅, L+R₆, L+R₇, L+R₈, L+R₉, L+R₁₀, L+R₁₁, L+R₁₂, L+SiO^{er}₂, L+SiO^{er}₂, L+SiO^{er}₂, L+Al₂O₃, R₁+R₆, R₂+R₆, R₂+R₇, R₂+R₈, R₂+R₁₁, R₃+R₁₁, R₄+R₁₁, R₅+R₁₂, R₈+R₁₁, R₉+R₁₁, R₁₀+R₁₁, R₁₀+R₁₁, R₁₀+R₁₂, R₁₁+R₁₂, R₄+R₁₂, R₁₂+SiO₂, R₁₂+Al₂O₃) and 46 three-phase regions (L₁+L₂+SiO₂, L+CaO+R₁, L+CaO+R₆, L+R₁+R₂, L+R₁+R₆, L+R₂+R₃, L+R₂+R₆, L+R₂+R₇, L+R₂+R₈, L+R₂+R₁₁, L+R₃+R₄, L+R₃+R₁₁, L+R₄+R₁₁, L+R₄+R₁₂, L+R₄+R₁₀, L+R₅+SiO^a₂, L+R₅+SiO^a₂, L+R₅+Al₂O₃, L+R₆+R₇, L+R₇+R₈, L+R₈+R₉, L+R₈+R₁₁, L+R₉+R₁₀, L+R₉+R₁₁, L+R₁₀+R₁₁, L+R₁₀+R₁₁, L+R₁₀+R₁₁, R₂+R₆+R₇, R₂+R₇+R₈, R₂+R₈+R₁₁, R₃+R₄+R₁₁, R₄+R₁₁+R₁₂, R₄+R₁₂+SiO₂, R₅+R₁₂+SiO₂, R₅+R₁₂+SiO₂, R₅+R₁₂+Al₂O₃, R₈+R₉+R₁₁, R₉+R₁₀+R₁₁, R₁₀+R₁₁, R₁₀+R₁₁, R₁₀+R₁₂+R₁₂+Al₂O₃).



Figure 2. Space model (a) and fragmentation (b) of surface S_{15} .

3. Crystallization paths

No more then 10 crystallization paths have been usually analyzed in the traditional tutorials for the system $CaO-Al_2O_3-SiO_2$ [2]. Computer model gives possible to consider the crystallization paths for all diagram concentration fields, 117 of which are two-dimensional, 163 - one-dimensional and 45 - zero-dimensional(figure 1b). Let' consider the compositions given in the CaO crystallization region.

So the melt G₁(0.722; 0.162; 0.116), arranged in the triangle R₁R₆CaO (figure 5a-b), moves along the ray CaO-G₁ (on the concentration projection) to the liquidus line p₁Q₁ while the passing through two-phase region L+CaO. Then it falls into three-phase region L+CaO+R1 and shifts along the line of monovariant equilibrium p₁Q₁. A reaction L(Q₂)+R₁ \rightarrow R₂+R₆ is finished with the deficit of melt on horizontal complex R₁Q₁R₆CaO at temperature Q₁ and beneath there are only crystals CaO, R₁ and R₆.

The melt G₂(0.638; 0.105; 0.257), given in triangle R₁R₂R₆ (figure c-d), moves along CaO-G to liquidus line p_1Q_1 at the going through two-phase region L+CaO. Next motion along monovariant equilibrium line p_1Q_1 to point Q₁ corresponds the three-phase region L+CaO+R1. Later the melt puts into three-phase region L+R1+R6 and its crystallization path coincides with the curve Q₁Q₂ to point Q₂. The reaction L(Q₂)+R₁ \rightarrow R₂+R₆ is compelled with the melt deficit on the horizontal complex R₁R₂Q₁R₆ at the temperature Q₂. Below this complex there are crystals R₁, R₂ and R₆.

Composition $G_3(0.706; 0.248; 0.046)$ also is given in subsystem $R_1R_2R_6$ but other concentration field (figure 5e). The melt G_3 shifts along segment CaO- G_2 to monovariant line p_1Q_1 within the region L+CaO and along the fragment of line p_1Q_1 within the region L+CaO+R1. Then the melt appears in region L+R1 and its crystallization paths across the liquidus surface R1($p_1e_1Q_2Q_1$) to line e_1Q_2 . The motion along line e_1Q_2 to point Q_2 corresponds the region L+R1+R2. The reaction L(Q_2)+R₁ \rightarrow R₂+R₆ ends with the melt deficit on the plane $R_1R_2Q_2R_6$ and beneath there are only crystals R_1 , R_2 and R_6 .

The melt $G_4(0.604; 0.017; 0.379)$, situated in triangle $R_2R_6R_7$ (figure 3f), moves along segment CaO-G₁ to point (1) on the liquidus line p_4Q_1 at the passing through two-phase region L+CaO (figure 3g). Next it falls into three-phase region L+CaO+R6 and shifts along line p_4Q_1 to point (2). Then melt moves the liquidus surface $p_4Q_1Q_2Q_3$ to point (3) on line Q_2Q_3 within region L+R6 and moves along

line Q_2Q_3 to point Q_3 at the going through L+R2+R6. The reaction $L(Q_3)+R_6 \rightarrow R_2+R_7$ is finished on horizontal complex $R_6R_2Q_3R_7$ and below there are only R_2 , R_6 and R_7 .



Figure 3. TXY models (a, c) and XY projections of melts trajectories G_1 (b), G_2 (d), G_3 (e) and G_4 (f, g is enlarged fragment).

4. Conclusions

Obtained CaO-Al₂O₃-SiO₂ model can be used as a template for the simulation of different silicate systems computer models (such as a system MgO-Al₂O₃-SiO₂ with similar but more simple topological structure). The crystallization stage for given composition also can be considered on the material balance diagrams. The elaborated phase diagrams computer models make possible to calculate the crystallization paths and microstructures formation at the silicate materials design.

References

- [1] Toropov N A, Bazarkovsky V P, Lapshin V V et al. 1972 *Diagram of Silicate System. Ternary Silicate Systems* vol 3 (Leningrad: Nauka) pp 184-190 (In Russian)
- [2] Patshenko A A, Aleksenko N V et al. 1977 *Phisical Chemistry of Silicate* ed A A Patshenko (Kiev: Vyshcha schkola) (In Russian)
- [3] Danek V 2006 Physico-chemical Analysis of Molten Electrolytes (New York: Elsevier)
- [4] Berman R G and Brown T H 1984 Geochim. et Cosmochim. Acta 48 661
- [5] Levin E M, Robbins C R and McMurdie H F 1964 *Phase Diagrams for Ceramists* (Ohio: American. Ceramic Society)
- [6] Shobu K 2009 CALPHAD 33 279
- [7] Lutsyk V I, Zyryanov A M and Zelenaya A E 2008 Russ. J. Inorg. Chem 53 792
- [8] Scheffe H 1957 J. Royal Statistical Soc 20 344
- [9] Lutsyk V I 1987 Analysis of Liquidus Surfaces in Ternary Systems (Moscow: Nauka Publ. House) (In Russian)