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# Melt solidification in the ceramic system $\mathrm{CaO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$ 

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#### Abstract

Technique of T-x-y diagram for system $\mathrm{CaO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$ computer simulation by the use of kinematical surfaces has been suggested. Original 245 crystallization paths were found.


## 1. Introduction

The system $\mathrm{CaO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}(\mathrm{C}-\mathrm{A}-\mathrm{S})$ has a great practical importance [1-2]. Its T-x-y diagram models [3-5] solve series of applied tasks, but don't allow to present full topological structure of diagram for the obtaining of concentration fields with the unique crystallization schemes. In particular, the immiscibility surfaces and 4 liquidus surfaces $\left(\mathrm{C}_{3} \mathrm{~S}, \mathrm{C}_{3} \mathrm{~S}_{2}, \mathrm{C}_{3} \mathrm{~A}, \mathrm{CA}_{6}\right)$ are absent in Vladimir Danek's model [3, P. 147]: "The immiscibility region near the $\mathrm{SiO}_{2}$ apex was neglected in the calculation as such behavior is not considered in the thermodynamic model. Furthermore, because of the lack of thermodynamic data, the crystallization of rankinite, tricalcium silicate, tricalcium aluminate, and calcium hexaaluminate was not included in the calculation". In the model, considered in [4], there isn't one binary eutectic in system C-A. As a result, a liquidus surface of $\mathrm{C}_{12} \mathrm{~A}_{7}$ is internal and, unlike the experimental data [5], doesn't adjacent to the binary system. The model calculated by the thermodynamics software CaTCalc [6] doesn't include three liquidus surfaces ( $\mathrm{C}_{3} \mathrm{~S}, \mathrm{C}_{3} \mathrm{~S}_{2}, \mathrm{C}_{12} \mathrm{~A}_{7}$ ).

The kinematical method is used for T-x-y diagram simulation [7], when a kinematical surface is produced by the motion of forming line along to the directing ones described by the interpolating polynomials. Approximating possibilities of the kinematical simulation were checked by the investigation of saddle surface (as the most complex boundary of phase regions) described by the Sheffe's model [8-9]. When kinematical saddle surface was simulated by 7 points ( 3 simplex vertexes, 3 points on its sides and $1-$ within the simplex), a standard deviation with the Sheffe's saddle (in 66 sites of concentration triangle with the step 0,1 ) was about 2,5 degrees and the most discordance (5-8 degrees) was observed in 17 points. The saddle points differ in the concentration coordinates: polynomial model $(0,181 ; 0.384 ; 0,435)$ and kinematical one $(0,159 ; 0.394 ; 0,447)$. When the kinematical surface was taken with 3 internal directing curves and simulated by 21 point ( 3 tops, 3 triads on the simplex' edges and 3 internal triads), the disagreements in temperatures were lowered and the concentration coordinates of saddle points for both models essentially approach each other: 0,$194 ; 0.391 ; 0,415$ - for a kinematical model.

## 2. Computer model of system $\mathbf{C a O}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$

The coordinates of 14 binary points ( 9 eutectics (e) and 5 peritectics (p)), 16 ternary points ( 6 eutectics (E), 9 quasiperitectics (Q) and the point of four-phase mass regrouping with two allotropies of $\mathrm{SiO}_{2}$ (U)), 10 binary ( $\mathrm{R}_{1}-\mathrm{R}_{10}$ ) and 2 ternary compounds $\left(\mathrm{R}_{11}\right.$ and $\left.\mathrm{R}_{12}\right)$ are taken for the computer model
construction $[1,7]$. Also the coordinates of inflections on monovariant liquidus curves should be taken into account. The following designation of compounds are used: $R_{1}-C_{3} S, R_{2}-C_{2} S, R_{3}-C_{3} S_{2}, R_{4}-$ $\mathrm{CS}, \mathrm{R}_{5}-\mathrm{A}_{3} \mathrm{~S}_{2}, \mathrm{R}_{6}-\mathrm{C}_{3} \mathrm{~A}, \mathrm{R}_{7}-\mathrm{C}_{12} \mathrm{~A}_{7}, \mathrm{R}_{8}-\mathrm{CA}, \mathrm{R}_{9}-\mathrm{CA}_{2}, \mathrm{R}_{10}-\mathrm{CA}_{6}, \mathrm{R}_{11}-\mathrm{C}_{2} \mathrm{AS}, \mathrm{R}_{12}-\mathrm{CAS}_{2}$, where $R_{1}, R_{3}, R_{6}, R_{10}$ are ingongruently melting compounds and $R_{2}, R_{4}, R_{5}, R_{7}, R_{8}, R_{9}, R_{11}, R_{12}$ are the congruently melting ones.

The phase diagram involves 16 unruled surfaces ( 15 liquidus surfaces S and one immiscibility surface i) (table 1, figure 1a), 80 ruled surfaces ( $77 \mathrm{~S}^{\mathrm{r}}+3 \mathrm{i}^{\mathrm{r}}$ ), 16 horizontal complexes at the invariant points temperatures $(\mathrm{H})$ and 16 vertical planes of triangulation (figure 1 b ).

Unruled surfaces with more then four points on their contour are constructed by the fragmentation into the more simple parts. In such way the surfaces $S_{2}, S_{4}, S_{6}, S_{7}, S_{11}, S_{14}, S_{16}$ are given by two parts, and the surfaces $S_{9}, S_{12}, S_{13}, S_{15}$ - by three parts. E.g., surface $\mathrm{S}_{15}$ has seven points on contour, the horizontal curve $\mathrm{p}_{3} \mathrm{U}$ and the cut mkn remained after intersection with immiscibility surface (figure 2 ). It includes 3 fragments $\mathrm{np}_{3} \mathrm{U}$, $\mathrm{mknUe}_{4}$ and $\mathrm{SiO}_{2} \mathrm{me}_{4}$, simulated by a common directing curve $p_{3}(n, U)\left(m, e_{4}\right) \mathrm{SiO}_{2}$ and by 2 forming lines $n(n, U) U$ and $m\left(m, e_{4}\right) e_{4}$. The parts of common directing line are used for the construction of different surface's fragments. So line's parts $p_{3}(n, U),(n, U)\left(m, e_{4}\right)$ and $\left(\mathrm{m}, \mathrm{e}_{4}\right) \mathrm{SiO}_{2}$ form the fragments $\mathrm{snp}_{3} \mathrm{U}, \mathrm{mknUe}_{4}$ and $\mathrm{SiO}_{2} \mathrm{me}_{4}$ accordingly. The cut is given as directing line $\mathrm{n}(\mathrm{n}, \mathrm{k}) \mathrm{k}(\mathrm{m}, \mathrm{k}) \mathrm{m}$ with five points on the intermediate surface's fragment. The intermediate point $\left(p_{3}, U\right)$ should be assigned at temperature $T_{U(p 3)}$ for the horizontal line $p_{3} U$ generation.

Table 1. Contours of unruled surfaces (figure 1a).

| Name | Contour | Name | Contour |
| :---: | :---: | :---: | :---: |
| i | $\mathrm{k}^{0} \mathrm{nkm}$ | $\mathrm{S}_{8}\left(\mathrm{C}_{3} \mathrm{~S}\right)$ | $\mathrm{p}_{1} \mathrm{Q}_{1} \mathrm{Q}_{2} \mathrm{e}_{1}$ |
| $\mathrm{S}_{1}(\mathrm{C})$ | $\mathrm{CaOp}_{1} \mathrm{Q}_{1} \mathrm{p}_{4}$ | $\mathrm{S}_{9}\left(\mathrm{C}_{2} \mathrm{~S}\right)$ | $\mathrm{e}_{1} \mathrm{Q}_{2} \mathrm{Q}_{3} \mathrm{E}_{1} \mathrm{Q}_{4}\left(\mathrm{Q}_{4}, \mathrm{Q}_{9}\right) \mathrm{Q}_{9} \mathrm{p}_{2} \mathrm{R}_{2}$ |
| $\mathrm{S}_{2}\left(\mathrm{C}_{3} \mathrm{~A}\right)$ | $\mathrm{p}_{4} \mathrm{Q}_{1} \mathrm{Q}_{2} \mathrm{Q}_{3} \mathrm{e}_{6}$ | $\mathrm{S}_{10}\left(\mathrm{C}_{3} \mathrm{~S}_{2}\right)$ | $\mathrm{p}_{2} \mathrm{Q}_{9} \mathrm{E}_{2} \mathrm{e}_{2}$ |
| $\mathrm{S}_{3}\left(\mathrm{C}_{12} \mathrm{~A}_{7}\right)$ | $\mathrm{e}_{6} \mathrm{Q}_{3} \mathrm{E}_{1} \mathrm{e}_{7} \mathrm{R}_{7}$ | $\mathrm{S}_{11}(\mathrm{CS})$ | $\mathrm{e}_{2} \mathrm{E}_{2}\left(\mathrm{E}_{2}, \mathrm{E}_{3}\right) \mathrm{E}_{3}\left(\mathrm{E}_{3}, \mathrm{E}_{4}\right) \mathrm{E}_{4} \mathrm{e}_{3} \mathrm{R}_{4}$ |
| $\mathrm{S}_{4}(\mathrm{CA})$ | $\mathrm{e}_{7} \mathrm{E}_{1} \mathrm{Q}_{4} \mathrm{Q}_{5} \mathrm{e}_{8} \mathrm{R}_{8}$ | $\mathrm{S}_{12}\left(\mathrm{C}_{2} \mathrm{AS}\right)$ | $\mathrm{Q}_{4} \mathrm{Q}_{5}\left(\mathrm{Q}_{5}, \mathrm{Q}_{6}\right) \mathrm{Q}_{6} \mathrm{E}_{6} \mathrm{E}_{3}\left(\mathrm{E}_{2}, \mathrm{E}_{3}\right) \mathrm{E}_{2} \mathrm{Q}_{9}$ |
| $\mathrm{S}_{5}\left(\mathrm{CA}_{2}\right)$ | $\mathrm{e}_{8} \mathrm{Q}_{5}\left(\mathrm{Q}_{5}, \mathrm{Q}_{6}\right) \mathrm{Q}_{6} \mathrm{e}_{9}$ | $\mathrm{S}_{13}\left(\mathrm{CAS}_{2}\right)$ | $\mathrm{E}_{3}\left(\mathrm{E}_{3}, \mathrm{E}_{4}\right) \mathrm{E}_{4}\left(\mathrm{E}_{4}, \mathrm{E}_{5}\right) \mathrm{E}_{5} \mathrm{Q}_{8}\left(\mathrm{Q}_{7}, \mathrm{Q}_{8}\right) \mathrm{Q}_{7} \mathrm{E}_{6}$ |
| $\mathrm{S}_{6}\left(\mathrm{CA}_{6}\right)$ | $\mathrm{e}_{9} \mathrm{Q}_{6} \mathrm{E}_{6} \mathrm{Q}_{7} \mathrm{p}_{5}$ | $\mathrm{S}_{14}\left(\mathrm{~S}^{\text {t }}\right.$ ) | $\mathrm{e}_{3} \mathrm{E}_{4}\left(\mathrm{E}_{4}, \mathrm{E}_{5}\right) \mathrm{E}_{5} \mathrm{Up}_{3}$ |
| $\mathrm{S}_{7}(\mathrm{~A})$ | $\mathrm{Al}_{2} \mathrm{O}_{3} \mathrm{p}_{5} \mathrm{Q}_{7}\left(\mathrm{Q}_{7}, \mathrm{Q}_{8}\right) \mathrm{Q}_{8} \mathrm{e}_{5}$ | $\begin{aligned} & \mathrm{S}_{15}\left(\mathrm{~S}^{\mathrm{c}}\right) \\ & \mathrm{S}_{16}\left(\mathrm{~A}_{3} \mathrm{~S}_{2}\right) \end{aligned}$ | $\mathrm{SiO}_{2} \mathrm{mknp}_{3} \mathrm{Ue}_{4}$ $\mathrm{e}_{4} \mathrm{UE}_{5} \mathrm{Q}_{8} \mathrm{e}_{5} \mathrm{R}_{5}$ |



Figure 1. XY projections of liquidus surfaces and immiscibility cupola (a), all diagram surfaces (b).
16 invariant transformations characterize the considered system. There are nine quasiperitectic transformations $(\mathrm{Q})$, six eutectic ones ( E ) and one transformation of the mass regrouping with two allotropies of $\mathrm{SiO}_{2} \quad(\mathrm{U}): \quad \mathrm{L}\left(\mathrm{Q}_{8}\right)+\mathrm{Al}_{2} \mathrm{O}_{3} \rightarrow \mathrm{R}_{5}+\mathrm{R}_{12}, \quad \mathrm{~L}\left(\mathrm{Q}_{5}\right)+\mathrm{R}_{9} \rightarrow \mathrm{R}_{8}+\mathrm{R}_{11}, \quad \mathrm{~L}\left(\mathrm{Q}_{7}\right)+\mathrm{Al}_{2} \mathrm{O}_{3} \rightarrow \mathrm{R}_{10}+\mathrm{R}_{12}$, $\mathrm{L}\left(\mathrm{Q}_{6}\right)+\mathrm{R}_{9} \rightarrow \mathrm{R}_{10}+\mathrm{R}_{11}, \quad \mathrm{~L}\left(\mathrm{Q}_{1}\right)+\mathrm{CaO} \rightarrow \mathrm{R}_{1}+\mathrm{R}_{6}, \quad \mathrm{U}+\mathrm{SiO}_{2}{ }_{2} \rightarrow \mathrm{R}_{5}+\mathrm{SiO}^{\mathrm{cr}}, \quad \mathrm{L}\left(\mathrm{Q}_{2}\right)+\mathrm{R}_{1} \rightarrow \mathrm{R}_{2}+\mathrm{R}_{6}$,

$$
\mathrm{L}\left(\mathrm{E}_{6}\right) \rightarrow \mathrm{R}_{10}+\mathrm{R}_{11}+\mathrm{R}_{12}, \quad \mathrm{~L}\left(\mathrm{Q}_{4}\right)+\mathrm{R}_{11} \rightarrow \mathrm{R}_{2}+\mathrm{R}_{8}, \quad \mathrm{~L}\left(\mathrm{Q}_{3}\right)+\mathrm{R}_{6} \rightarrow \mathrm{R}_{2}+\mathrm{R}_{7}, \quad \mathrm{~L}\left(\mathrm{E}_{5}\right) \rightarrow \mathrm{R}_{5}+\mathrm{R}_{12}+\mathrm{SiO}_{2}
$$

$$
\mathrm{L}\left(\mathrm{E}_{1}\right) \rightarrow \mathrm{R}_{2}+\mathrm{R}_{7}+\mathrm{R}_{8}, \mathrm{~L}\left(\mathrm{Q}_{9}\right)+\mathrm{R}_{2} \rightarrow \mathrm{R}_{3}+\mathrm{R}_{11}, \mathrm{~L}\left(\mathrm{E}_{2}\right) \rightarrow \mathrm{R}_{3}+\mathrm{R}_{4}+\mathrm{R}_{11}, \mathrm{~L}\left(\mathrm{E}_{3}\right) \rightarrow \mathrm{R}_{4}+\mathrm{R}_{11}+\mathrm{R}_{12}, \mathrm{~L}\left(\mathrm{E}_{4}\right) \rightarrow \mathrm{R}_{4}+\mathrm{R}_{12}+\mathrm{SiO}_{2}
$$

Phase diagram includes one one-phase region (L), 33 two-phase regions $\left(\mathrm{L}_{1}+\mathrm{L}_{2}, \mathrm{~L}+\mathrm{CaO}, \mathrm{L}+\mathrm{R}_{1}\right.$, $\mathrm{L}+\mathrm{R}_{2}, \mathrm{~L}+\mathrm{R}_{3}, \mathrm{~L}+\mathrm{R}_{4}, \mathrm{~L}+\mathrm{R}_{5}, \mathrm{~L}+\mathrm{R}_{6}, \mathrm{~L}+\mathrm{R}_{7}, \mathrm{~L}+\mathrm{R}_{8}, \mathrm{~L}+\mathrm{R}_{9}, \mathrm{~L}+\mathrm{R}_{10}, \mathrm{~L}+\mathrm{R}_{11}, \mathrm{~L}+\mathrm{R}_{12}, \mathrm{~L}+\mathrm{SiO}^{\mathrm{cr}}{ }_{2}, \mathrm{~L}+\mathrm{SiO}_{2}{ }_{2}^{\mathrm{tr}}$, $\mathrm{L}+\mathrm{Al}_{2} \mathrm{O}_{3}, \mathrm{R}_{1}+\mathrm{R}_{6}, \mathrm{R}_{2}+\mathrm{R}_{6}, \mathrm{R}_{2}+\mathrm{R}_{7}, \mathrm{R}_{2}+\mathrm{R}_{8}, \mathrm{R}_{2}+\mathrm{R}_{11}, \mathrm{R}_{3}+\mathrm{R}_{11}, \mathrm{R}_{4}+\mathrm{R}_{11}, \mathrm{R}_{5}+\mathrm{R}_{12}, \mathrm{R}_{8}+\mathrm{R}_{11}, \mathrm{R}_{9}+\mathrm{R}_{11}, \mathrm{R}_{10}+\mathrm{R}_{11}$, $\left.\mathrm{R}_{10}+\mathrm{R}_{12}, \quad \mathrm{R}_{11}+\mathrm{R}_{12}, \quad \mathrm{R}_{4}+\mathrm{R}_{12}, \quad \mathrm{R}_{12}+\mathrm{SiO}_{2}, \quad \mathrm{R}_{12}+\mathrm{Al}_{2} \mathrm{O}_{3}\right)$ and 46 three-phase regions $\left(\mathrm{L}_{1}+\mathrm{L}_{2}+\mathrm{SiO}_{2}\right.$, $\mathrm{L}+\mathrm{CaO}+\mathrm{R}_{1}, \mathrm{~L}+\mathrm{CaO}+\mathrm{R}_{6}, \mathrm{~L}+\mathrm{R}_{1}+\mathrm{R}_{2}, \mathrm{~L}+\mathrm{R}_{1}+\mathrm{R}_{6}, \mathrm{~L}+\mathrm{R}_{2}+\mathrm{R}_{3}, \mathrm{~L}+\mathrm{R}_{2}+\mathrm{R}_{6}, \mathrm{~L}+\mathrm{R}_{2}+\mathrm{R}_{7}, \mathrm{~L}+\mathrm{R}_{2}+\mathrm{R}_{8}, \mathrm{~L}+\mathrm{R}_{2}+\mathrm{R}_{11}$, $\mathrm{L}+\mathrm{R}_{3}+\mathrm{R}_{4}, \mathrm{~L}+\mathrm{R}_{3}+\mathrm{R}_{11}, \mathrm{~L}+\mathrm{R}_{4}+\mathrm{R}_{11}, \mathrm{~L}+\mathrm{R}_{4}+\mathrm{R}_{12}, \mathrm{~L}+\mathrm{R}_{4}+\mathrm{SiO}^{\mathrm{cr}}, \mathrm{L}+\mathrm{R}_{5}+\mathrm{R}_{12}, \mathrm{~L}+\mathrm{R}_{5}+\mathrm{SiO}_{2}^{\mathrm{u}}, \mathrm{L}+\mathrm{R}_{5}+\mathrm{SiO}_{2}^{\mathrm{d}}$, $\mathrm{L}+\mathrm{R}_{5}+\mathrm{Al}_{2} \mathrm{O}_{3}, \quad \mathrm{~L}+\mathrm{R}_{6}+\mathrm{R}_{7}, \quad \mathrm{~L}+\mathrm{R}_{7}+\mathrm{R}_{8}, \quad \mathrm{~L}+\mathrm{R}_{8}+\mathrm{R}_{9}, \quad \mathrm{~L}+\mathrm{R}_{8}+\mathrm{R}_{11}, \quad \mathrm{~L}+\mathrm{R}_{9}+\mathrm{R}_{10}, \quad \mathrm{~L}+\mathrm{R}_{9}+\mathrm{R}_{11}, \quad \mathrm{~L}+\mathrm{R}_{10}+\mathrm{R}_{11}$, $\mathrm{L}+\mathrm{R}_{10}+\mathrm{R}_{12}, \quad \mathrm{~L}+\mathrm{R}_{11}+\mathrm{R}_{12}, \quad \mathrm{~L}+\mathrm{R}_{10}+\mathrm{Al}_{2} \mathrm{O}_{3}, \quad \mathrm{~L}+\mathrm{R}_{12}+\mathrm{Al}_{2} \mathrm{O}_{3}, \quad \mathrm{~L}+\mathrm{R}_{12}+\mathrm{SiO}_{2}, \quad \mathrm{CaO}+\mathrm{R}_{1}+\mathrm{R}_{6}, \quad \mathrm{R}_{1}+\mathrm{R}_{2}+\mathrm{R}_{6}$, $\mathrm{R}_{2}+\mathrm{R}_{3}+\mathrm{R}_{11}, \mathrm{R}_{2}+\mathrm{R}_{6}+\mathrm{R}_{7}, \mathrm{R}_{2}+\mathrm{R}_{7}+\mathrm{R}_{8}, \mathrm{R}_{2}+\mathrm{R}_{8}+\mathrm{R}_{11}, \mathrm{R}_{3}+\mathrm{R}_{4}+\mathrm{R}_{11}, \mathrm{R}_{4}+\mathrm{R}_{11}+\mathrm{R}_{12}, \mathrm{R}_{4}+\mathrm{R}_{12}+\mathrm{SiO}_{2}, \mathrm{R}_{5}+\mathrm{R}_{12}+\mathrm{SiO}_{2}$, $\left.\mathrm{R}_{5}+\mathrm{R}_{12}+\mathrm{Al}_{2} \mathrm{O}_{3}, \mathrm{R}_{8}+\mathrm{R}_{9}+\mathrm{R}_{11}, \mathrm{R}_{9}+\mathrm{R}_{10}+\mathrm{R}_{11}, \mathrm{R}_{10}+\mathrm{R}_{11}+\mathrm{R}_{12}, \mathrm{R}_{10}+\mathrm{R}_{12}+\mathrm{Al}_{2} \mathrm{O}_{3}\right)$.


Figure 2. Space model (a) and fragmentation (b) of surface $S_{15}$.

## 3. Crystallization paths

No more then 10 crystallization paths have been usually analyzed in the traditional tutorials for the system $\mathrm{CaO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$ [2]. Computer model gives possible to consider the crystallization paths for all diagram concentration fields, 117 of which are two-dimensional, 163 - one-dimensional and 45-zero-dimensional(figure 1 b ). Let' consider the compositions given in the CaO crystallization region.

So the melt $\mathrm{G}_{1}(0.722 ; 0.162 ; 0.116)$, arranged in the triangle $\mathrm{R}_{1} \mathrm{R}_{6} \mathrm{CaO}$ (figure 5a-b), moves along the ray $\mathrm{CaO}-\mathrm{G}_{1}$ (on the concentration projection) to the liquidus line $\mathrm{p}_{1} \mathrm{Q}_{1}$ while the passing through two-phase region $\mathrm{L}+\mathrm{CaO}$. Then it falls into three-phase region $\mathrm{L}+\mathrm{CaO}+\mathrm{R} 1$ and shifts along the line of monovariant equilibrium $p_{1} Q_{1}$. A reaction $L\left(Q_{2}\right)+R_{1} \rightarrow R_{2}+R_{6}$ is finished with the deficit of melt on horizontal complex $\mathrm{R}_{1} \mathrm{Q}_{1} \mathrm{R}_{6} \mathrm{CaO}$ at temperature $\mathrm{Q}_{1}$ and beneath there are only crystals $\mathrm{CaO}, \mathrm{R}_{1}$ and $\mathrm{R}_{6}$.

The melt $\mathrm{G}_{2}\left(0.638 ; 0.105 ; 0.257\right.$ ), given in triangle $\mathrm{R}_{1} \mathrm{R}_{2} \mathrm{R}_{6}$ (figure $\mathrm{c}-\mathrm{d}$ ), moves along CaO-G to liquidus line $\mathrm{p}_{1} \mathrm{Q}_{1}$ at the going through two-phase region $\mathrm{L}+\mathrm{CaO}$. Next motion along monovariant equilibrium line $\mathrm{p}_{1} \mathrm{Q}_{1}$ to point $\mathrm{Q}_{1}$ corresponds the three-phase region $\mathrm{L}+\mathrm{CaO}+\mathrm{R} 1$. Later the melt puts into three-phase region $L+R 1+R 6$ and its crystallization path coincides with the curve $Q_{1} Q_{2}$ to point $\mathrm{Q}_{2}$. The reaction $\mathrm{L}\left(\mathrm{Q}_{2}\right)+\mathrm{R}_{1} \rightarrow \mathrm{R}_{2}+\mathrm{R}_{6}$ is compelled with the melt deficit on the horizontal complex $R_{1} R_{2} Q_{1} R_{6}$ at the temperature $Q_{2}$. Below this complex there are crystals $R_{1}, R_{2}$ and $R_{6}$.

Composition $G_{3}(0.706 ; 0.248 ; 0.046)$ also is given in subsystem $R_{1} R_{2} R_{6}$ but other concentration field (figure 5e). The melt $\mathrm{G}_{3}$ shifts along segment $\mathrm{CaO}-\mathrm{G}_{2}$ to monovariant line $\mathrm{p}_{1} \mathrm{Q}_{1}$ within the region $\mathrm{L}+\mathrm{CaO}$ and along the fragment of line $\mathrm{p}_{1} \mathrm{Q}_{1}$ within the region $\mathrm{L}+\mathrm{CaO}+\mathrm{R} 1$. Then the melt appears in region $L+R 1$ and its crystallization paths across the liquidus surface $R 1\left(p_{1} e_{1} Q_{2} Q_{1}\right)$ to line $e_{1} Q_{2}$. The motion along line $e_{1} Q_{2}$ to point $Q_{2}$ corresponds the region $L+R 1+R 2$. The reaction $L\left(Q_{2}\right)+R_{1} \rightarrow R_{2}+R_{6}$ ends with the melt deficit on the plane $\mathrm{R}_{1} \mathrm{R}_{2} \mathrm{Q}_{2} \mathrm{R}_{6}$ and beneath there are only crystals $\mathrm{R}_{1}, \mathrm{R}_{2}$ and $\mathrm{R}_{6}$.

The melt $G_{4}(0.604 ; 0.017 ; 0.379)$, situated in triangle $R_{2} R_{6} R_{7}$ (figure 3 f ), moves along segment $\mathrm{CaO}-\mathrm{G}_{1}$ to point (1) on the liquidus line $\mathrm{p}_{4} \mathrm{Q}_{1}$ at the passing through two-phase region $\mathrm{L}+\mathrm{CaO}$ (figure 3 g ). Next it falls into three-phase region $\mathrm{L}+\mathrm{CaO}+\mathrm{R} 6$ and shifts along line $\mathrm{p}_{4} \mathrm{Q}_{1}$ to point (2). Then melt moves the liquidus surface $\mathrm{p}_{4} \mathrm{Q}_{1} \mathrm{Q}_{2} \mathrm{Q}_{3}$ to point (3) on line $\mathrm{Q}_{2} \mathrm{Q}_{3}$ within region $\mathrm{L}+\mathrm{R} 6$ and moves along

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line $Q_{2} Q_{3}$ to point $Q_{3}$ at the going through $L+R 2+R 6$. The reaction $L\left(Q_{3}\right)+R_{6} \rightarrow R_{2}+R_{7}$ is finished on horizontal complex $\mathrm{R}_{6} \mathrm{R}_{2} \mathrm{Q}_{3} \mathrm{R}_{7}$ and below there are only $\mathrm{R}_{2}, \mathrm{R}_{6}$ and $\mathrm{R}_{7}$.


Figure 3. TXY models $(a, c)$ and $X Y$ projections of melts trajectories $G_{1}(b), G_{2}(d), G_{3}(e)$ and $G_{4}(f, g$ is enlarged fragment).

## 4. Conclusions

Obtained $\mathrm{CaO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$ model can be used as a template for the simulation of different silicate systems computer models (such as a system $\mathrm{MgO}-\mathrm{Al}_{2} \mathrm{O}_{3}-\mathrm{SiO}_{2}$ with similar but more simple topological structure). The crystallization stage for given composition also can be considered on the material balance diagrams. The elaborated phase diagrams computer models make possible to calculate the crystallization paths and microstructures formation at the silicate materials design.

## References

[1] Toropov N A, Bazarkovsky V P, Lapshin V V et al. 1972 Diagram of Silicate System. Ternary Silicate Systems vol 3 (Leningrad: Nauka) pp 184-190 (In Russian)
[2] Patshenko A A, Aleksenko N V et al. 1977 Phisical Chemistry of Silicate ed A A Patshenko (Kiev: Vyshcha schkola) (In Russian)
[3] Danek V 2006 Physico-chemical Analysis of Molten Electrolytes (New York: Elsevier)
[4] Berman R G and Brown T H 1984 Geochim. et Cosmochim. Acta 48661
[5] Levin E M, Robbins C R and McMurdie H F 1964 Phase Diagrams for Ceramists (Ohio: American. Ceramic Society)
[6] Shobu K 2009 CALPHAD 33279
[7] Lutsyk V I, Zyryanov A M and Zelenaya A E 2008 Russ. J. Inorg. Chem 53792
[8] Scheffe H 1957 J. Royal Statistical Soc 20344
[9] Lutsyk V I 1987 Analysis of Liquidus Surfaces in Ternary Systems (Moscow: Nauka Publ. House) (In Russian)

