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2008 Supercond. Sci. Technol. 21 082001

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## RAPID COMMUNICATION

# Superconductivity at 53.5 K in $\text{GdFeAsO}_{1-\delta}$

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Received 3 May 2008, in final form 5 May 2008

Published 20 May 2008

Online at [stacks.iop.org/SUST/21/082001](http://stacks.iop.org/SUST/21/082001)

## Abstract

Here we report the fabrication and superconductivity of the iron-based arsenic oxide  $\text{GdFeAsO}_{1-\delta}$  compound with oxygen-deficiency, which has an onset resistivity transition temperature at 53.5 K. This material has the same crystal structure as the newly discovered high- $T_c$   $\text{ReFeAsO}_{1-\delta}$  family (Re = rare earth metal) and a further reduced crystal lattice, while the  $T_c$  starts to decrease compared with the  $\text{SmFeAsO}_{1-\delta}$  system.

(Some figures in this article are in colour only in the electronic version)

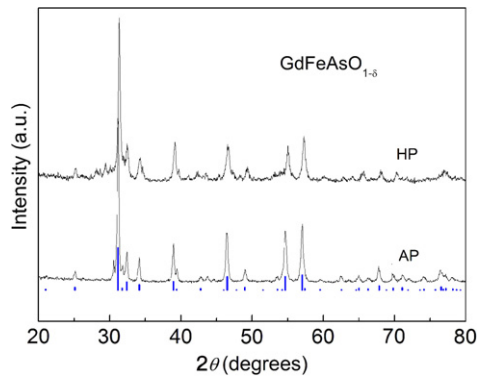
Since the discovery of layered copper oxide superconductors [1], extensive research has been performed to understand the mechanism of high temperature superconductivity and to explore higher  $T_c$  materials. It is believed that strong electron correlation and layered structures may play an important role. However, the  $T_c$  of the non-copper-based superconductors is still lower than 40 K as predicted by BCS theory and experimental evidence. Recently, superconductivity in iron- and nickel-based layered quaternary compounds has been reported:  $\text{LaOFeP}$  ( $T_c \sim 4$  K) [2],  $\text{LaONiP}$  ( $T_c \sim 3$  K) [3], then  $\text{LaOFeAs}$  at 26 K with substitution of As for P and F-doping at the oxygen site [4]. These discoveries have attracted much interest in further experiments and theoretical studies. Subsequently, the  $T_c$  rapidly increases to above 50 K in the  $\text{ReFeAsO}_{1-x}\text{F}_x$  family [5–9], with the replacement of La by other light rare-earth elements such as Pr, Nd, Sm, etc. Instead of F-doping, we have recently succeeded in synthesizing the  $\text{ReFeAsO}_{1-\delta}$  superconductors, which have better superconducting properties [10]. These quaternary superconductors crystallize with the tetragonal layered  $\text{ZrCuSiAs}$  structure, in the space group of  $P4/nmm$ , which has a structure of alternating Fe–As layers and Re–O layers, similar to that

of cuprates. As for the Re = Gd system, we have discovered its superconductivity previously in some multi-phased samples. Here we report the superconductivity in  $\text{GdFeAsO}_{1-\delta}$  and  $\text{GdFeAsO}_{1-x}\text{F}_x$  systems.

A series of superconductors with nominal compositions of oxygen-deficient  $\text{GdFeAsO}_{1-\delta}$  and F-doped  $\text{GdFeAsO}_{1-x}\text{F}_x$  were prepared by the high pressure (HP) synthesis method. The starting chemicals Gd chips, As, Fe,  $\text{Fe}_2\text{O}_3$ , and  $\text{FeF}_2$  powders are all with purity better than 99.99%. At the first step, GdAs powder was obtained by reacting Gd pieces and As powders at 650 °C for 12 h and then 1150 °C for 12 h. The starting materials were mixed together according to the nominal ratio, then ground thoroughly and pressed into small pellets. The pellets were sealed in boron nitride crucibles and sintered in high-pressure synthesis apparatus under a pressure of 6 GPa and temperature of 1350 °C for 2 h. The HP samples are hard and can be polished to a shiny mirror-surface.

The phase purity and structural identification were characterized by powder x-ray diffraction (XRD) analysis on an MXP18A-HF type diffractometer with  $\text{Cu K}\alpha$  radiation from 20° to 80° with a step of 0.01°. The XRD results indicate that the main phase of all F-doped and oxygen-deficient samples adopts the same  $\text{ReFeAsO}$  structure with slight impurity phases. The impurity phases have been

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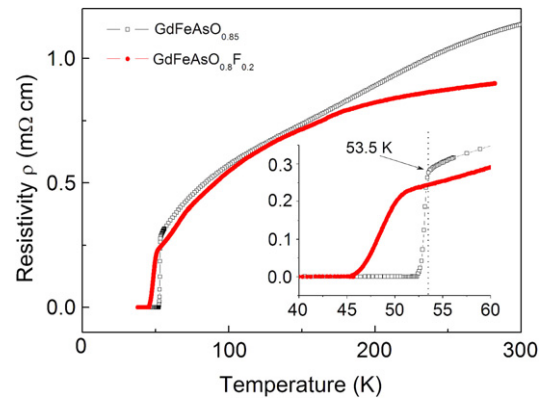


**Figure 1.** Typical XRD patterns for the nominal  $\text{GdFeAsO}_{0.85}$  sample synthesized by the HP method (upper line) and  $\text{GdFeAsO}$  by the AP method (lower line); the vertical bars indicate the calculated diffraction peaks for the  $\text{GdFeAsO}$ .

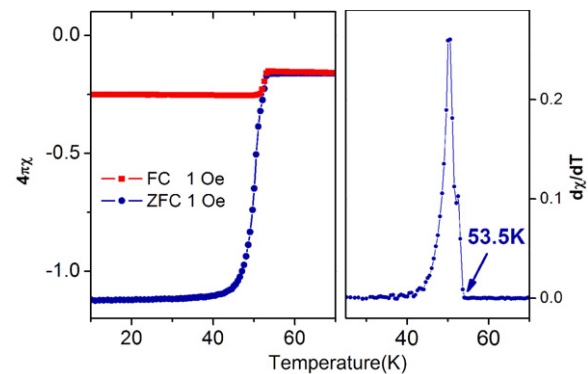
identified to be some by-products and do not superconduct at the measuring temperature. Figure 1 shows the comparison of a typical XRD pattern for a nominal  $\text{GdFeAsO}_{0.85}$  sample synthesized by the HP method and an ambient-pressure (AP) synthesized undoped  $\text{GdFeAsO}$  sample. The lattice parameters are  $a = 3.890(4)$  Å,  $c = 8.383(2)$  Å for the HP sample of  $\text{GdFeAsO}_{0.85}$ , and  $a = 3.903(7)$  Å,  $c = 8.453(1)$  Å for an AP sample of  $\text{GdFeAsO}$ , which indicates a clear shrinkage for the oxygen-deficient sample compared with the undoped sample. All samples with either F-doping or oxygen-deficiency have shrunken crystal lattices. The lattice parameters of a Gd-system are smaller than those of an Sm-system reported in our previous paper [10], which shows a further enhanced chemical pressure on the Fe–As plane in the Gd-system.

The resistivity of all samples was measured by the standard four-probe method from 4 to 300 K. Slight F-doping and oxygen-deficiency both lead to the occurrence of superconductivity in this system, while samples with oxygen-deficiency were found to have higher  $T_c$  compared with F-doped ones, and all samples have metallic behavior up to 300 K. The sample with a nominal composition of  $\text{GdFeAsO}_{0.85}$  was found to have the highest  $T_c$  in this Gd-system, while for the F-doping system, the nominal  $\text{GdFeAsO}_{0.8}\text{F}_{0.2}$  was found to have the highest  $T_c$ , and the corresponding resistivity curves are shown in figure 2. It can be seen in the inset that a clear superconducting onset transition ( $T_c$  (onset)) occurred at 53.5 K and a zero resistivity transition ( $T_c$  (zero)) at 52.3 K for the  $\text{GdFeAsO}_{0.85}$ ; while for the  $\text{GdFeAsO}_{0.8}\text{F}_{0.2}$ , the  $T_c$  (onset) and  $T_c$  (zero) are at 51.2 and 45.5 K. Compared with the  $\text{SmFeAsO}_{1-\delta}$  superconductor, the  $T_c$  starts to decrease, which may indicate that the  $T_c$  has reached a maximum in the Sm-based system by the inner chemical pressure effect for the  $\text{ReFeAsO}_{1-\delta}$  family.

The DC magnetization was measured using a Quantum Design MPMS XL-1 system. For an experiment cycle the sample was cooled to 1.8 K in zero field cooling (ZFC) and data were gathered when warming in an applied field, then cooled again under an applied field (FC) and measured when warming up. The DC-susceptibility data (measured under a magnetic field of 1 Oe) of the  $\text{GdFeAsO}_{0.85}$  are



**Figure 2.** The temperature dependence of resistivity for the nominal composition of  $\text{GdFeAsO}_{0.85}$  and  $\text{GdFeAsO}_{0.8}\text{F}_{0.2}$  synthesized by the HP method.



**Figure 3.** The temperature dependence of the DC-susceptibility and the differential ZFC curve for the nominal  $\text{GdFeAsO}_{0.85}$  sample synthesized by the HP method.

shown in figure 3, with the differential ZFC curve in the right panel. The sharp magnetic transitions on DC curves indicate the good quality of this superconducting component. The onset diamagnetic superconducting transition temperature determined from the differential ZFC curve is 53.5 K, the same as the onset transition point on the corresponding resistivity curve.

In conclusion, we have succeeded in preparing the  $\text{GdFeAsO}$  superconductors by both F-doping and oxygen-deficiency. The oxygen-deficient samples were found to have better superconducting properties in this system and the highest  $T_c$  is at 53.5 K for the nominal  $\text{GdFeAsO}_{0.85}$  composition.

## Acknowledgments

We thank Mrs Shun-Lian Jia for her kind help in resistivity measurements. This work is supported by the Natural Science Foundation of China (NSFC, Nos 50571111 and 10734120) and the 973 program of China (Nos 2006CB601001 and 2007CB925002). We also acknowledge support from the EC under the project COMEPHS TTC.

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